

Superconductivity in semimetallic $\text{Bi}_3\text{O}_2\text{S}_3$

L. Li,¹ D. Parker,¹ P. Babkevich,² L. Yang,^{2,3} H. M. Ronnow,² and A. S. Sefat¹

¹*Materials Science & Technology Division, Oak Ridge National Laboratory, Oak Ridge, Tennessee 37831, USA*

²*Laboratory for Quantum Magnetism (LQM), École Polytechnique Fédérale de Lausanne (EPFL), CH-1015 Lausanne, Switzerland*

³*Laboratory of Physics of Complex Matter, École Polytechnique Fédérale de Lausanne (EPFL), CH-1015 Lausanne, Switzerland*

(Received 9 December 2014; revised manuscript received 1 February 2015; published 12 March 2015)

Here we report a further investigation on the thermodynamic and transport properties, and an assessment of theoretical calculations, for the BiS_2 -layered $\text{Bi}_3\text{O}_2\text{S}_3$ superconductor. The polycrystalline sample is synthesized with a superconducting transition temperature of $T_c^{\text{onset}} = 5.75$ K and $T_c^{\text{zero}} = 4.03$ K ($\approx T_c^{\text{mag}}$) that drops to 3.3 K by applying a hydrostatic pressure of 6 kbar. Density-of-states (DOS) calculations give substantial hybridization between Bi, O, and S, with Bi the largest component of DOS, which supports the idea that the BiS_2 layer is relevant for producing electron-phonon coupling. An analysis of previously published specific heat data for $\text{Bi}_3\text{O}_2\text{S}_3$ is additionally suggestive of a strong electron-phonon interaction in the Bi-O-S system. The analysis of the Seebeck coefficient results strongly suggests that $\text{Bi}_3\text{O}_2\text{S}_3$ is a semimetal. In fact, the semimetallic or narrow band gap behavior may occur in certain other materials in the BiS_2 -layered class of materials, such as $\text{Bi}_4\text{O}_4\text{S}_3$.

DOI: [10.1103/PhysRevB.91.104511](https://doi.org/10.1103/PhysRevB.91.104511)

PACS number(s): 74.20.Pq, 74.25.fg

I. INTRODUCTION

Understanding the reasons for superconductivity in Bi-O-S systems is important, especially since high-temperature superconductivity is seen in other Bi-based materials, such as $\text{Ba}_{1-x}\text{K}_x\text{BiO}_3$ and $\text{Bi}_2\text{Sr}_2\text{Ca}_2\text{Cu}_3\text{O}_{10+x}$ [1]. The superconducting members of Bi-O-S systems are based on BiS_2 layers with large unit cells composed of various spacer layers. The members include $\text{Bi}_4\text{O}_4\text{S}_3$ with $T_c = 4.4$ K, $H_{c2(0)} \sim 3$ T, and a dominant-electron concentration of $\sim 2 \times 10^{19} \text{ cm}^{-3}$ at 300 K [2–4]. The enhanced density of states (DOS) at the Fermi level from Bi $6p_x$ and $6p_y$, and the strong Fermi surface nesting of quasi-one-dimensional bands are predicted to play a role in Cooper pairing [2]. Theoretical tight-binding calculations on a similar BiS_2 -layer-based system of LaOBiS_2 (F doping; $T_c = 10.6$ K) [5] shows that the dominating bands for electron conduction and superconductivity are those derived from four Bi $6p-S$ $3p$ orbitals that give nearly half-filled, electron-hole symmetric, quasi-one-dimensional bands [6] that give nesting of the Fermi surface. Other members of Bi-O-S systems include $\text{Bi}_6\text{O}_8\text{S}_5$ [7] and $\text{Bi}_3\text{O}_2\text{S}_3$ [8,9]. Most recently, a critical temperature at $T_c = 4.5$ K was discovered in $\text{Bi}_3\text{O}_2\text{S}_3$ [8] that appeared, based on that work, to be a typical type-II superconductor with $T_c^{\text{onset}} = 5.8$ K, an upper critical field of $H_{c2} = 4.8$ T, and low electron-dominant carriers of $\sim 10^{19} \text{ cm}^{-3}$ [9]. This paper focuses on $\text{Bi}_3\text{O}_2\text{S}_3$ by advancing investigations into the reasons for superconductivity through theoretical calculations and bulk thermodynamic and transport property results. A schematic image of the unit cell is shown in the right inset of Fig. 1(a), depicting layers of $(\text{BiS}_2)_2$ (rocksalt type), Bi_2O_2 (fluorite type), and S_2 . We find that T_c decreases by an application of 6 kbar, similar to $\text{Bi}_4\text{O}_4\text{S}_3$ [10]. From Seebeck coefficient results and calculations, we find that $\text{Bi}_3\text{O}_2\text{S}_3$ is in fact a semimetal and the band overlap is most likely in the range of 27–41 meV. From a study of previously published specific heat data for $\text{Bi}_3\text{O}_2\text{S}_3$, demonstrating low Fermi-level DOS, we also find evidence of a strong electron-phonon interaction in Bi-O-S systems.

II. EXPERIMENT

A polycrystalline sample of $\text{Bi}_3\text{O}_2\text{S}_3$ was prepared by a conventional solid state reaction method according to Refs. [8,9]. The Bi_2O_3 powders and S pieces were weighed by targeting “ $\text{Bi}_{12}\text{O}_{18}\text{S}_{13}$,” fully mixed, ground, and then pressed into pellets. The pellets were sealed in an evacuated silica tube and placed in a prewarmed furnace at 430 °C. After 4 h, the sample was quenched in ice water. The sample was removed from the tube, reground and repressed, and then sealed in another quartz tube. The tube was placed back in the preheated furnace at 430 °C. After 14 h, the sample was quenched down to 0 °C with ice water. After another regrinding and pelletizing, the pellets were sealed in an evacuated silica tube and placed in a preheated furnace at 520 °C for 17 h, and then quenched again. The resultant sample appeared black and dense. The x-ray diffraction (XRD) pattern was collected by a PANalytical X’Pert PRO MPD x-ray powder diffractometer with $\text{CuK}\alpha$ radiation. The Rietveld refinement of the XRD patterns was done with the FULLPROF software suite, using the reported Wyckoff positions [8] and the $I4/mmm$ space group. The electrical and thermal transport measurements were performed in a Quantum Design (QD) physical property measurement system (PPMS), using a standard four-probe method. Magnetization measurements were performed in a QD magnetic property measurement system (MPMS). The application of pressure was performed in an Mcell 10 pressure cell (easyLab Technologies) inside the MPMS; the applied pressure was estimated from the T_c of the Pb manometer. First-principles calculations were performed using the linearized augmented plane-wave code WIEN2K [11], employing the generalized gradient approximation (GGA) of Perdew, Burke, and Ernzerhof [12]. RK_{max} was set to 9.0, where R is the smallest muffin-tin radius and K_{max} the largest plane-wave wave vector. Muffin-tin radii for all calculations, in units of the Bohr radius a_0 , were as follows: Bi, 2.31; O, 1.89; S, 2.08, except for the dimerized S, which was taken as 1.8. All internal coordinates were optimized until forces were less than 2 mRy/bohr, while lattice parameters

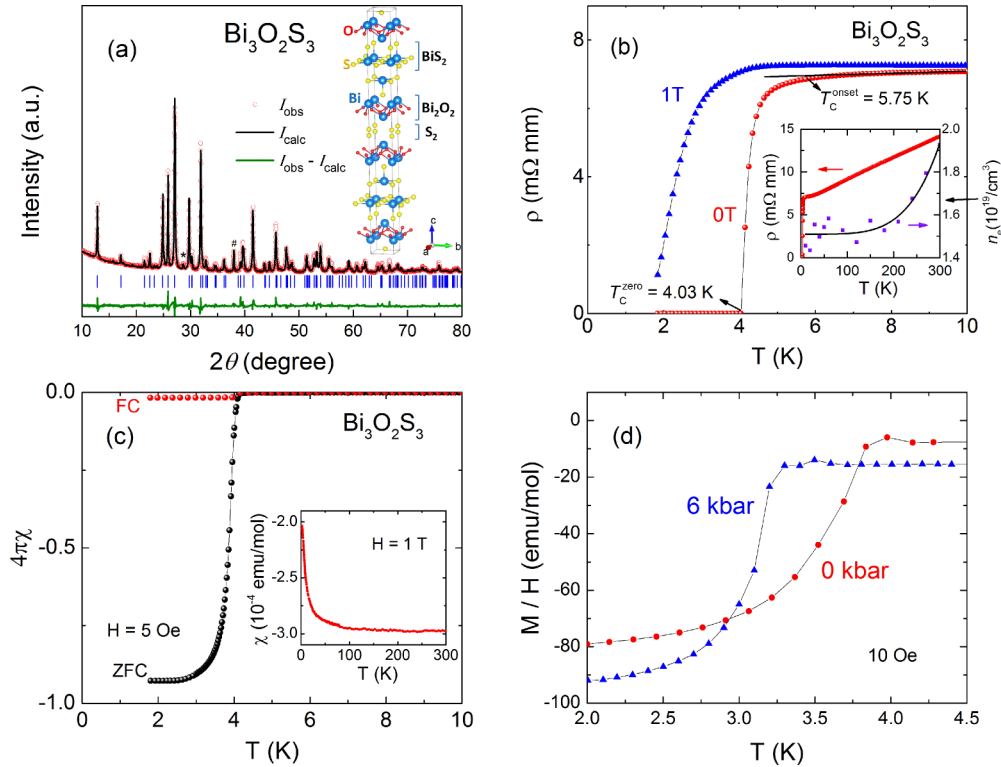


FIG. 1. (Color online) (a) Powder x-ray diffraction pattern for $\text{Bi}_3\text{O}_2\text{S}_3$. Red circles represent observed data; black and green solid lines represent the calculated intensity and difference between the observed and calculated intensity; blue vertical bars indicate the Bragg reflection positions for $\text{Bi}_3\text{O}_2\text{S}_3$. * and # indicate the peaks of Bi_2S_3 and Bi, respectively. The inset shows the schematic image of the crystal structure of $\text{Bi}_3\text{O}_2\text{S}_3$. (b) Temperature dependence of resistivity for $\text{Bi}_3\text{O}_2\text{S}_3$ measured below 10 K, at 0 and 1 T applied magnetic fields. The inset is the temperature dependence of resistivity and charge carrier concentration for $\text{Bi}_3\text{O}_2\text{S}_3$ up to room temperature at 0 T. (c) Temperature dependence of magnetic susceptibility for $\text{Bi}_3\text{O}_2\text{S}_3$ measured below 10 K, at $H = 5$ Oe applied magnetic fields. The inset displays the temperature dependence of magnetic susceptibility for $\text{Bi}_3\text{O}_2\text{S}_3$ up to room temperature obtained at 1 T applied magnetic fields. (d) Temperature dependence of magnetic susceptibility for $\text{Bi}_3\text{O}_2\text{S}_3$ measured below 4.5 K, with applying a hydrostatic pressure of 0 kbar (red circle) and 6 kbar (blue triangle), respectively.

were taken from experiment. Here, 1000 or more k points in the full Brillouin zone were used for all calculations. Spin-orbit coupling was included for all calculations, except for the optimization. We also performed Boltzmann transport calculations of the thermopower using the BOLTZTRAP code [13]. For these calculations as many as 40 000 k points in the full Brillouin zone were used.

III. RESULTS AND DISCUSSION

Figure 1(a) shows the XRD pattern for polycrystalline $\text{Bi}_3\text{O}_2\text{S}_3$, which looks very similar to that reported in Ref. [8]. Only tiny impurity amounts of Bi_2OS_2 [8], Bi_2S_3 , and Bi are present, which have also been taken into account for a refinement that improved the fitting. The refined lattice constants are $a = 3.9645(5)$ Å and $c = 41.3165(4)$ Å. Impurity phases were estimated using the Hill and Howard method [14], and their total contents were less than 8% by mass.

The temperature dependence of resistivity is shown in Fig. 1(b). In zero field, $T_c^{\text{onset}} = 5.75$ K and $T_c^{\text{zero}} = 4.03$ K. Upon application of a 1 T magnetic field it can be seen that the T_c^{onset} shifts to lower temperature and the sample does not reach zero resistivity down to 2 K. The inset of Fig. 1(b) shows resistivity up to room temperature. It displays

a typical metallic behavior, similar to that of $\text{Bi}_4\text{O}_4\text{S}_3$ [2]. The temperature dependence of magnetic susceptibility is shown in Fig. 1(c). The sharp transition in $\chi(T)$ indicates that the sample is rather homogeneous. The value of T_c , defined by the divergence of zero-field-cooled and field-cooled results, is about 4.1 K. The shielding volume fraction at 1.8 K is $\sim 93\%$. Higher field measurements [the inset of Fig. 1(c)] reveal that $\text{Bi}_3\text{O}_2\text{S}_3$ exhibits a diamagnetic behavior without any abrupt drop or anomaly, indicating there is no magnetic ordering in this system. Compressing the structural lattice parameters by applying external pressure is believed to be an effective method for exploring the optimized condition for superconductivity. To this end, we have performed a pressure study of the temperature dependence of susceptibility for $\text{Bi}_3\text{O}_2\text{S}_3$ below 5 K. The hydrostatic pressure does not improve the superconductivity of the sample, while, on the contrary, the transition temperature $T_c^{\text{mag}} \sim 4$ K at ambient pressure decreases to 3.3 K at 6 kbar, as shown in Fig. 1(d). Although a similar T_c suppression was seen in $\text{Bi}_4\text{O}_4\text{S}_3$ [10], further theoretical work on the DOS and a structural analysis at high pressure could be useful for a better understanding of this effect.

The band structure calculation of $\text{Bi}_3\text{O}_2\text{S}_3$ is shown in Fig. 2(a); it suggests that this material is a semiconductor with a very small band gap (approximately 25 meV) and substantial

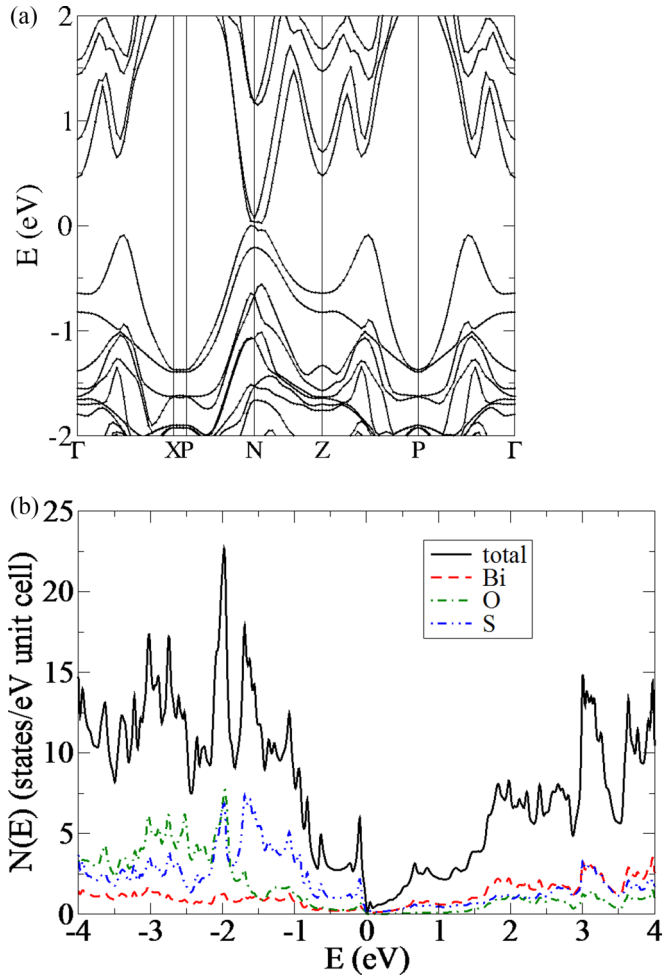


FIG. 2. (Color online) (a) The calculated band structure of $\text{Bi}_3\text{O}_2\text{S}_3$, within the body-centered-tetragonal Brillouin zone. The energy zero is set to the valence band maximum. Note the very small band gap—approximately 25 meV—at the N point. (b) The calculated density of states of $\text{Bi}_3\text{O}_2\text{S}_3$. The energy 0 is set to the valence band maximum.

band degeneracy, particularly in the conduction bands, where a second band minimum occurs less than 50 meV above the primary band minimum, with all these extrema at or very near the N point. The calculated density of states for this material is shown in Fig. 2(b). The plot shows a comparatively heavy band just below the valence band maximum, and a somewhat lighter band in the conduction band. This band shows substantial hybridization between the Bi, O, and S. Previous papers have argued for the importance of the BiS_2 layers in generating electron-phonon coupling [2], and this hybridization is supportive of this idea.

One natural interpretation of our experimental data in light of these plots is that $\text{Bi}_3\text{O}_2\text{S}_3$ is indeed a narrow band semiconductor, with an extrinsic n -type doping level of approximately $1.5 \times 10^{19} \text{ cm}^{-3}$, as suggested by the Hall data within a single-sign conduction model. However, one immediately finds two problems with this explanation. First, this inferred carrier concentration is the same as found in the prior work (see Ref. [9]), and were this simply a semiconductor, such a coincidence in extrinsic doping levels would seem rather

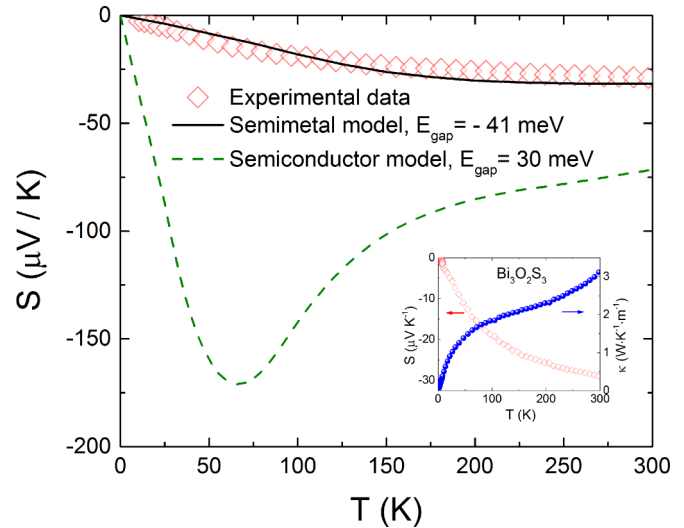


FIG. 3. (Color online) The measured thermopower (diamonds) and calculated thermopower under two scenarios: dashed line, the semiconducting band structure; solid line, an assumed semimetallic band structure with a band overlap of 41 meV. The inset shows the temperature dependence of the Seebeck coefficient and thermal conductivity up to room temperature.

unlikely. Second, and more compellingly, the thermopower data presented in Fig. 3 shows strong evidence of semimetallic, rather than semiconducting, behavior. In Fig. 3 we plot the experimental Seebeck data, along with two theoretical fits to the data. These fits were generated using the first-principles calculated band structure, with the thermopower calculated via the Boltzmann transport code BOLTZTRAP [13], using the constant scattering time approximation, which has shown considerable success in describing doped semiconductors. Note that the thermopower was calculated from the transport results as the conductivity-weighted thermopower [i.e., $S = (\sigma_{xx}S_x + \sigma_{yy}S_y + \sigma_{zz}S_z)/(\sigma_{xx} + \sigma_{yy} + \sigma_{zz})$], which is appropriate for a polycrystalline sample; please see Ref. [15] for details of this procedure.

The first fit (dashed line) was plotted using the first-principles calculated band gap of 30 meV for a chemical carrier concentration of $n = 1.2 \times 10^{19} \text{ cm}^{-3}$, comparatively near the value that would be obtained from the Hall number assuming single-sign conduction (note that these two need not be exactly equal). This fit shows a low-temperature thermopower slope S/T ($T < 50 \text{ K}$), approximately *ten* times the measured value, and then rapidly enters the bipolar regime soon thereafter. Neither of these behaviors resembles the gently increasing curve in the experimental data (we suspect this data would show a maximum at $T \sim 350 \text{ K}$). We have also tried fitting the data with a much heavier chemical doping ($n = 1.5 \times 10^{20} \text{ cm}^{-3}$, not shown). While one can fit the initial slope of $S(T)$ in this way, the thermopower magnitude at temperatures above 50 K continues to increase linearly, reaching magnitudes much larger than the experimental value. This is not surprising considering that $E_F(T = 0)$ in this case is approximately 150 meV above the conduction band minimum, and nearly 180 meV above the valence band maximum, so that bipolar effects are minimal in this temperature range at this

doping. The second curve (black solid line) was generated using the first-principles calculated band structure with a “scissor shift” employed. In this “scissor shift” the conduction and valence band structures are taken as unchanged, but the band gap is adjusted. The plot indicates that a band overlap of $0.003 \text{ Ry} = 41 \text{ meV}$ (i.e., a -41 meV band gap) produces a good fit to the experimental data. We take this as strong evidence that $\text{Bi}_3\text{O}_2\text{S}_3$ is in fact a semimetal, not a semiconductor. Note that we are also able to fit the data with a somewhat smaller band overlap of 27 meV , so we can say that the actual band overlap is most likely in the range of $27\text{--}41 \text{ meV}$. This also implies that the actual carrier concentration likely differs significantly from the value the Hall number would indicate, if a single-sign conduction model were assumed.

We note that this apparent overstatement of the band gap in the electronic structure calculations is unusual, given that the GGA usually understates band gaps due to a self-interaction error and other issues. However, it is currently an outstanding electronic structure problem to compute band gaps sufficiently accurately to accurately predict room temperature transport properties, and the problem is only exacerbated at temperatures below room temperature. While progress has been made [16] in calculating band gaps, one sometimes is simply forced to resort to so-called “scissor shifts,” as implemented here and in Ref. [17] to attain quantitative agreement with experiment. We do not have a definite explanation for the band gap overstatement, but note that band gap understatement is not a necessary consequence of all such first-principles calculations. For example, in Ref. [18] the band gap of FeSi is calculated as approximately 0.1 eV while transport measurements (in this case, resistivity) indicate a gap of approximately 0.05 eV . In that work it is asserted that FeSi is strongly correlated, and there have been similar suggestions in the literature [19] regarding possible correlation effects in the oxybismuth sulfides. We think it possible that the neglect of correlation effects in our density-functional theory calculations accounts for the apparent overstatement of the band gap.

With the electronic structure established, let us consider the superconductivity itself in more detail. Let us now make an approximate estimate of the strength of the electron-phonon interaction that is presumed responsible for the superconductivity, using present and previous data. In Ref. [9] a value of the T -linear specific heat constant $\gamma = 1.65 \text{ mJ/mol K}^2$ was reported. On a density-of-states basis this is $0.7/\text{eV}$ formula unit, or $0.23/\text{eV}$ BiS unit, noting that previous works [6] have argued the BiS_2 layers to be the relevant component producing electron-phonon coupling, the presumed pairing source.

Assuming a modified BCS expression $T_c \sim \omega_D \exp[-(1 + \lambda)/\lambda]$, including the Eliashberg mass renormalization, and neglecting the Coulomb pseudopotential μ^* , a $5 \text{ K } T_c$ with the 188 K Debye temperature estimated in Ref. [9] corresponds to an electron-phonon coupling constant $\lambda = N_0V$ of 0.38 . This yields a modified electron-phonon interaction potential V of 1.5 eV per BiS unit. This rough calculation suggests a particularly strong interaction and one that could presumably be strengthened significantly via charge doping to increase N_0 , if feasible, leading to larger T_c .

We note that the same features suggestive of semimetallic behavior in $\text{Bi}_3\text{O}_2\text{S}_3$ —the increasing thermal conductivity above the Debye temperature and likely saturation and turning over of the temperature dependent thermopower—appear to be present in the related material $\text{Bi}_4\text{O}_4\text{S}_3$ [20]. In this case we cannot easily distinguish between a semimetal and narrow band gap semiconductor, but note that both cases are strongly at odds with the band structure calculations for this material published in Ref. [2], which depict a band edge some 0.6 eV above E_F , i.e., metallic behavior. Hence the observance of semimetallic or narrow band gap behavior may be more common in this class of materials than previously thought.

IV. CONCLUSIONS

In summary, we have synthesized and confirmed the bulk superconductivity by both resistivity and magnetic susceptibility measurements in the BiS_2 -layer-based superconductor $\text{Bi}_3\text{O}_2\text{S}_3$. Hall results find the same Hall number as in previous work, and this, combined with a theoretical analysis of our Seebeck coefficient data, lead to the conclusion that $\text{Bi}_3\text{O}_2\text{S}_3$ is in fact a semimetal with a band overlap in the range of $27\text{--}41 \text{ meV}$. An analysis of previous data showing a low Fermi-level density of states [9] suggests that the electron-phonon interaction in this material is quite strong and that T_c could be increased by appropriate charge doping.

ACKNOWLEDGMENTS

This work was primarily supported by the US Department of Energy, Office of Science, Basic Energy Sciences, Materials Science and Engineering Division, and partially by ORNL’s LDRD funding (all theoretical modeling and calculations). The pressure measurements were funded by the Swiss National Science Foundation, its Sinergia Network Mott Physics Beyond the Heisenberg Model, and the European Research Council project CONQUEST; in addition, Jonathan White is thanked for his assistance.

-
- [1] R. J. Cava, B. Batlogg, J. J. Krajewski, R. Farrow, L. W. Rupp, Jr., A. E. White, K. Short, W. F. Peck, and T. Kometani, *Nature (London)* **332**, 814 (1988); C. Michel, M. Hervieu, M. M. Borel, A. Grandin, F. Deslandes, J. Provost, and B. Raveau, *Z. Phys. B* **68**, 421 (1987).
- [2] Y. Mizuguchi, H. Fujihisa, Y. Gotoh, K. Suzuki, H. Usui, K. Kuroki, S. Demura, Y. Takano, H. Izawa, and O. Miura, *Phys. Rev. B* **86**, 220510(R) (2012).
- [3] S. K. Singh, A. Kumar, B. Gahtori, S. Kirtan, G. Sharma, S. Patnaik, and V. P. S. Awana, *J. Am. Chem. Soc.* **134**, 16504 (2012).
- [4] P. Srivastava, Shruti, and S. Patnaik, *Supercond. Sci. Technol.* **27**, 055001 (2014).
- [5] Y. Mizuguchi, S. Demura, K. Deguchi, Y. Takano, H. Fujihisa, Y. Gotoh, H. Izawa, and O. Miura, *J. Phys. Soc. Jpn.* **81**, 114725 (2012).

- [6] H. Usui, K. Suzuki, and K. Kuroki, *Phys. Rev. B* **86**, 220501 (2012).
- [7] Y. Yu, J. Shao, S. Tan, C. Zhang, and Y. Zhang, *J. Phys. Soc. Jpn.* **82**, 034718 (2013).
- [8] W. A. Phelan, D. C. Wallace, K. E. Arpino, J. R. Neilson, K. J. Livi, C. R. Seabourne, A. J. Scott, and T. M. McQueen, *J. Am. Chem. Soc.* **135**, 5372 (2013).
- [9] J. Shao, Z. Liu, X. Yao, L. Pi, S. Tan, C. Zhang, and Y. Zhang, *Phys. Status Solidi RRL* **8**, 845 (2014).
- [10] H. Kotegawa, Y. Tomita, H. Tou, H. Izawa, Y. Mizuguchi, O. Miura, S. Demura, K. Deguchi, and Y. Takano, *J. Phys. Soc. Jpn.* **81**, 103702 (2012).
- [11] P. Blaha, K. Schwarz, G. K. H. Madsen, D. Kvasnicka, and J. Luitz, *WIEN2k, An Augmented Plane Wave + Local Orbitals Program for Calculating Crystal Properties* (Karlheinz Schwarz, Techn. Universität Wien, Austria, 2001).
- [12] J. P. Perdew, K. Burke, and M. Ernzerhof, *Phys. Rev. Lett.* **77**, 3865 (1996).
- [13] G. K. H. Madsen, K. Schwarz, P. Blaha, and D. J. Singh, *Phys. Rev. B* **68**, 125212 (2003).
- [14] R. J. Hill and C. J. Howard, *J. Appl. Crystallogr.* **20**, 467 (1987).
- [15] X. Chen, D. Parker, and D. J. Singh, *Sci. Rep.* **3**, 3168 (2013).
- [16] F. Tran and P. Blaha, *Phys. Rev. Lett.* **102**, 226401 (2009).
- [17] L. Lykke, B. B. Iversen, and G. K. H. Madsen, *Phys. Rev. B* **73**, 195121 (2006).
- [18] C. Fu, M. P. C. M. Krijn, and S. Doniach, *Phys. Rev. B* **49**, 2219 (1994).
- [19] Y. Liang, X. Wu, W. F. Tsai, and J. Hu, *Front. Phys.* **9**, 194 (2014).
- [20] S. G. Tan, L. J. Li, Y. Liu, P. Tong, B. C. Zhao, W. J. Lu, and Y. P. Sun, *Physica C* **483**, 94 (2012).