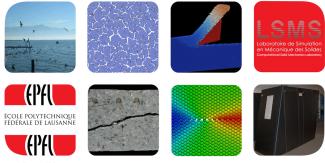
Linking Discrete Dislocations and Molecular Dynamics in 3D: a Start

T. Junge G. Anciaux W.A. Curtin J.-F. Molinari T. Nogaret



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Outline

MD modelling of friction Advances through atomic force microscopy MD scratching - simplest case MD size restrictions

CADD

Overview Coupling Scheme 3D Difficulties

Test case

Simplest case: no FEM required Simplest template Preliminary results

Outline

MD modelling of friction Advances through atomic force microscopy MD scratching - simplest case MD size restrictions

CADD

Test case



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Motivation

Friction is complex and poorly understood

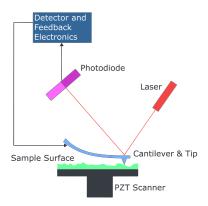


Questions

- Role of plasticity in friction?
- Proportion of W ending up as $W_{\rm pl}$?



Advances through atomic force microscopy



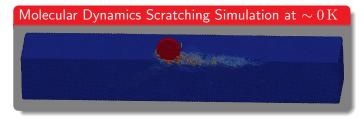
Atomic scale measurements

Source: Wikipedia

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MD scratching - simplest case

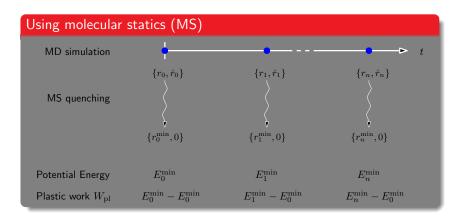


Advantages

- very few a priori assumptions
- deep understanding because of complete knowledge of each atom in the simulation box
- Dislocation nucleation and motion handled accurately

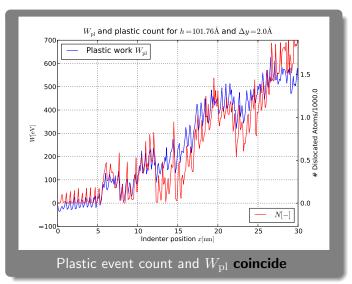


MD scratching - Computation of plastic work $W_{\rm pl}$





MD size restrictions - small indentation, simulation size sufficient



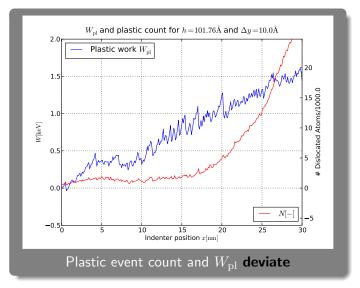


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MD size restrictions - deeper indentation, too small simulation box

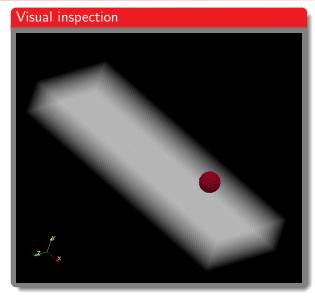




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MD size restrictions



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What to do?

Molecular dynamics simulations



Source: Ziegenhain et al. / J. Mech. Phys. Solids 57 (2009)

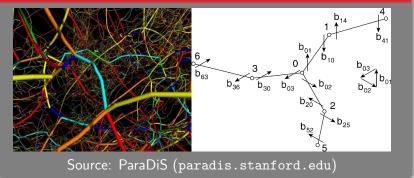
Accurate dislocation nucleation, but too small scale



EPH

What to do?

Discrete dislocation dynamics simulations

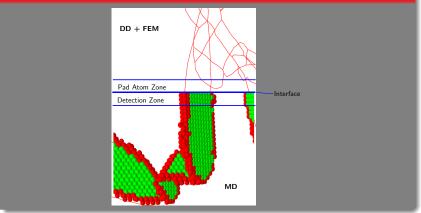


Larger scale, but problematic dislocation nucleation



A combined approach

Coupled atomistics and discrete dislocations (CADD)



Pioneered by Shilkrot, Curtin / J. Mech. Phys. Solids 50 (2002) (in 2D)

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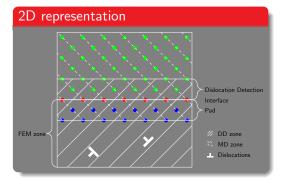
CADD Overview Coupling Scheme 3D Difficulties

Test case



CADD

Overview



Shilkrot, Curtin / J. Mech. Phys. Solids 50 (2002)

- 1. Interface atoms (red) are FEM boundary conditions
- 2. Pad atoms (blue) are MD boundary conditions
- DD are driven by FEM stress in continuum and by detection in atomistics

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EPEL







MD

- DD
- 1. move atoms
- 2. detect disls \rightarrow DD
- 1. move dislocations
- 2. adjust pad atoms \rightarrow MD

FEM

- 1. move elements
- 2. adjust pad atoms \rightarrow MD

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MD

- 1. move atoms
- 2. detect disls \rightarrow DD
- 3. set interface \rightarrow FEM

DD

- 1. move dislocations
- 2. adjust pad atoms \rightarrow MD
- 3. set image forces \rightarrow FEM

FEM

- 1. move elements
- 2. adjust pad atoms \rightarrow MD
- 3. set stresses \rightarrow DD

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MD

- 1. move atoms
- 2. detect disls \rightarrow DD
- 3. set interface \rightarrow FEM
- 4. return to 1

DD

- 1. move dislocations
- 2. adjust pad atoms \rightarrow MD
- 3. set image forces \rightarrow FEM
- 4. return to 1

FEM

- 1. move elements
- 2. adjust pad atoms \rightarrow MD
- 3. set stresses \rightarrow DD

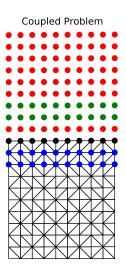
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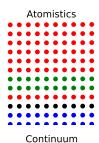
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 $4. \ \ \text{return to} \ 1$



CADD CADD in 2D





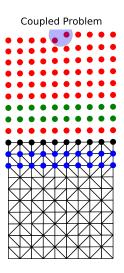


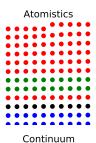
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CADD CADD in 2D





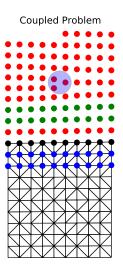


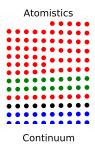
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CADD in 2D





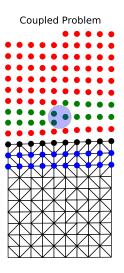


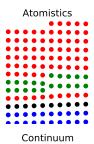
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CADD CADD in 2D



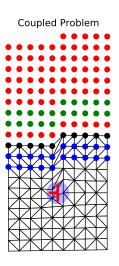


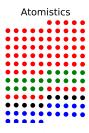


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CADD CADD in 2D

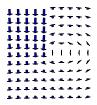




Continuum



+ Discrete Dislocation





CADD Simplifications obsolete in 3D

2D Dislocations are point entities
 Dislocations always in either the MD or the FEM+DD zone

 2D Dislocations never touch the interface High energy (and non-linear) cores don't interfere with interface

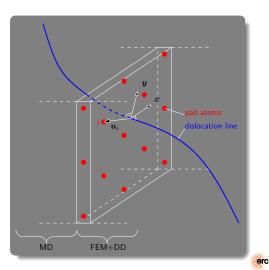


CADD Simplifications obsolete in 3D

- 2D Dislocations are point entities
 Dislocations always in either the MD or the FEM+DD zone
- **3D Dislocations are loops**Dislocation may be partly in the MD and the FEM+DD zone
- 2D Dislocations never touch the interface High energy (and non-linear) cores don't interfere with interface
- **!!** 3D Dislocations can cross the interface

For each Dislocation line

Apply displacement $oldsymbol{u}_i = oldsymbol{u}(x - x_{ ext{core}}, y - y_{ ext{core}})$ to each pad atom i

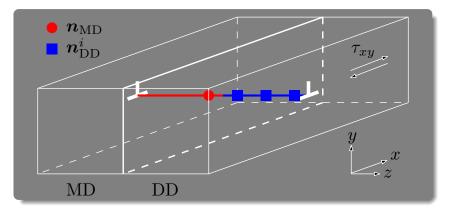


CADD

Test case Simplest case: no FEM required Simplest template Preliminary results



Simplest case: no FEM required



Simplification

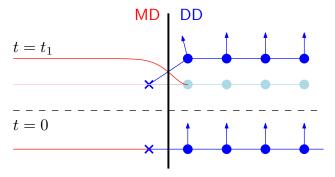
No out-of-plane displacement \Rightarrow no need for solving elasticity



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Revisit coupling scheme





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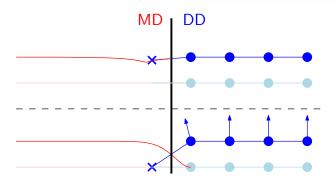
Revisit coupling scheme

MD

- 1. move atoms
- 2. detect dislocations \rightarrow DD

DD

- 1. move dislocations
- 2. adjust pad atoms \rightarrow MD



Revisit coupling scheme

MD

- 1. move atoms
- 2. detect dislocations \rightarrow DD
- 3. return to 1

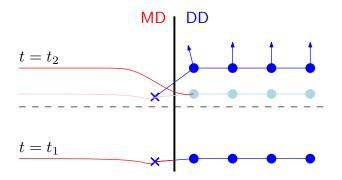
DD

- 1. move dislocations
- 2. adjust pad atoms \rightarrow MD

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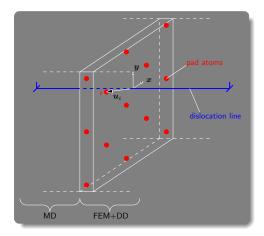
3. return to 1



Simplest case: simple template

$\boldsymbol{u}_z = 0$

Straight edge dislocation \perp interface The template is a 2D **Problem**



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Simplest template

isotropic linear elastic displacement field

$$\begin{split} u_x(\bar{x},\bar{y}) &= \frac{b}{2\pi} \left(\arctan \frac{\bar{x}}{\bar{y}} + \frac{\bar{x}\bar{y}}{2(1-\nu)(\bar{x}^2 + \bar{y}^2)} \right), \\ u_y(\bar{x},\bar{y}) &= -\frac{b}{2\pi} \left(\frac{1-2\nu}{1(1-\nu)} \ln(\bar{x}^2 + \bar{y}^2) + \frac{\bar{x}^2 - \bar{y}^2}{4(1-\nu)(\bar{x}^2 + \bar{y}^2)} \right), \\ \text{where } \bar{x} &= \frac{x}{b}, \quad \bar{y} = \frac{y}{b} \end{split}$$

Trivial core tracking

The slip plane of the dislocation is known!

- 1. discretise the detection zone in bins
- 2. compute for each bin $\Delta u_x = \max(u_x) \min(u_x)$
- 3. check for which bin Δu_x is closest to b/2



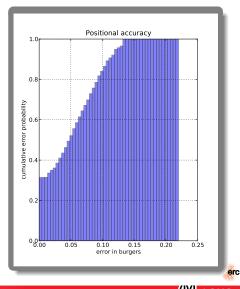
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Accuracy of the template

Material sytem

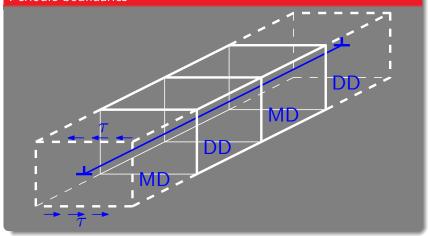
- Magnesium (hcp, MEAM
 [1])
- Prismatic edge dislocation
- Very compact core
- Good accuracy of linear elastic approximation

 M.I. Mendelev, M. Asta, M.J. Rahman and J.J. Hoyt, Phil. Mag. 89, 3269-3285 (2009)



Preliminary results- setup

Periodic boundaries

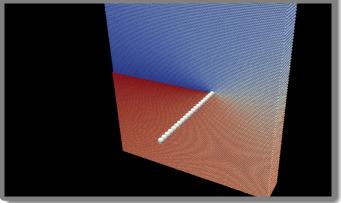




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Preliminary results

A sample movie





Preliminary results

Observations

- Detection and pad work
 The dislocation information is accurately communicated between the domains
- Dislocation line stays straight
 Same dislocation velocity in MD and DD
- No apparent artificial pinning The elastic solution is a good enough template for this case



Outlook - FEM coupling

$\mathsf{FEM} \Leftarrow \mathsf{MD}$

Interface atoms serve as FEM displacement boundary condition

$\mathsf{FEM} \Rightarrow \mathsf{MD}$

FEM computes **elastic** contribution to pad atom displacement

$\mathsf{FEM} \Leftarrow \mathsf{DD}$

Image forces serve as FEM traction boundary condition

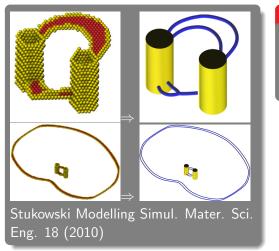
$\mathsf{FEM} \Rightarrow \mathsf{DD}$

FEM computes stresses on DD node positions



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Outlook - Dislocation detection



Stukowski 2010

Method based on burgers circuits, parallel, on the fly implementation.



Conclusions

1.

MD insufficient to treat friction problems of meaningful size



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Conclusions

 $\mathsf{MD}\xspace$ insufficient to treat friction problems of meaningful size





Conclusions

MD insufficient to treat friction problems of meaningful size

The 3D CADD method is functional

3.

2.

The 3D CADD method is a promising candidate to work out the size problem in atomic scale contact.



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Conclusions

MD insufficient to treat friction problems of meaningful size

The 3D CADD method is functional

3.

2.

The 3D CADD method is a promising candidate to work out the size problem in atomic scale contact.

4.

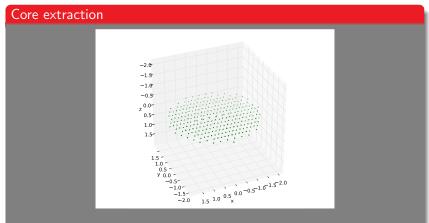
CADD also applicable to many other problems such as

- Fracture, crack propagation
- Dislocation interaction with grain boundaries

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Appendix – Core templates

Using real core data



- generate a dislocation core (MS)
- filter out low energy atoms

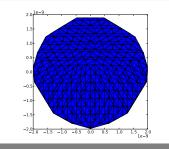
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Appendix – Core templates

Using real core data

Generating a core mesh



- Triangulate core using projected atoms as nodes
- compute displacement and interpolation shape functions S_u and S_x for each element:

$$oldsymbol{u}(oldsymbol{x}) = oldsymbol{S}_u \; S_x^{-1} \; oldsymbol{x}$$



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